(ANL-IN-02-081)

U.S. PATENT APPLICATION

for

LAYERED Li $_{1+X}$ Ni $_{\alpha}$ Mn $_{\beta}$ A $_{\gamma}$ O $_{2}$ CATHODE MATERIALS FOR Li-ION RECHARGEABLE BATTERIES AND METHODS OF MANUFACTURING THE SAME

Inventors: Sun-Ho KANG

Khalil AMINE

Attorneys:

FOLEY & LARDNER LLP 330 North Wabash Avenue, Suite 3300 Chicago IL 60611-3608 Telephone: 312.832.4500

Facsimile: 312.755.1925

(ANL-IN-02-081)

LAYERED $\text{Li}_{1+X}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{A}_{\gamma}\text{O}_{2}$ CATHODE MATERIALS FOR Li-ION RECHARGEABLE BATTERIES AND METHODS OF MANUFACTURING THE SAME

[0001] The United States Government has rights in this invention pursuant to Contract W-31-109-ENG-38 between the U.S. Department of Energy and The University of Chicago representing Argonne National Laboratory.

CROSS-REFERENCE TO RELATED PATENT APPLICATIONS

[0002] The present application is related to and claims priority from U.S. Prov. Appl. Ser. No. 60/454,930 which was filed on March 14, 2003 and titled "LAYERED Li_{1+X}Ni $_{\alpha}$ Mn $_{\beta}$ A $_{\gamma}$ O $_{2}$ CATHODE MATERIALS FOR Li-ION RECHARGEABLE BATTERIES AND METHODS OF MANUFACTURING THE SAME" and which is hereby incorporated by reference into the present application.

FIELD OF THE INVENTION

[0003] The present invention relates to a positive active material for lithium-ion rechargeable batteries. More particularly, the present invention relates to a positive active material based on layered lithium nickel-manganese oxide doped with various elements.

BACKGROUND OF THE INVENTION

[0004] Current lithium-ion batteries, or cells, use a solid reductant as the anode and a solid oxidant as the cathode. Solid-state, high energy-density batteries use metallic lithium as the anode. Lithium metal is a preferred electrode material as a result of its superior thermodynamic and kinetic properties. In addition, lithium is a good conductor of electricity and heat. Furthermore, lithium's malleability and ductility make it an excellent metal with which to work. However, lithium has well-known drawbacks when

(ANL-IN-02-081)

used as an anode material. For example, lithium is very reactive and often creating flammability concerns.

[0005] When the battery is discharged, the anode supplies Li⁺ ions to the Li⁺ ion electrolyte and electrons to the external circuit. The cathode is typically an electronically conducting host into which Li⁺ ions are inserted reversibly from the electrolyte as a guest species and are charge-compensated by electrons from the external circuit. Due to the unique properties of lithium, active materials have been developed for use in lithium-ion batteries. Primary batteries or cells are those in which the chemical reaction supplying the electrons is not reversible with respect to the closed universe of the battery. A secondary battery, or cell, utilizes a reaction which can be reversed when current is applied to the battery, thus "recharging" the battery. The chemical reactions at the anode and cathode of a lithium secondary battery must be reversible. On charge, the removal of electrons from the cathode by an external field releases Li⁺ ions back to the electrolyte to restore the parent host structure, and the addition of electrons to the anode by the external field attracts charge-compensating Li⁺ ions back into the anode to restore it to its original composition.

[0006] Li-Co-O and Li-Co-Ni-O based materials are currently utilized as cathodes in lithium-ion rechargeable batteries. Although useful, lithium-ion rechargeable batteries which include Li-Co-O and Li-Co-Ni-O based materials for cathodes include a number of drawbacks. In particular, the cobalt is very expensive, and the use of cobalt and nickel results in environmental issues and thermal safety concerns.

[0007] There is therefore a need for electrode materials that have improved properties and require less expensive materials.

SUMMARY OF THE INVENTION

[0008] The system of the present invention comprises a Li-Ni-Mn-A-O based material, where A may be a variety of metals, such as but not limited to Mg, Zn, Ga, B, Zr, Ti and Co. The A metals are utilized in very small amounts. These materials in their phase-pure

(ANL-IN-02-081)

state are good performers. However, during manufacture, as the systems are cooled after calcination, inactive phases develop which significantly lowers their performance. One advantageous feature of the present invention is utilizing a rapid quench directly after the high temperature calcination step to rapidly cool the layered oxides, thereby preventing growth of these undesirable phases.

[0009] It has been found that this technique works effectively for materials made by any of several cathode forming processes, such as for example but not limited to, a solid-state reaction method, an aqueous solution method, or a sol-gel method. When manufactured by such methods, layered cathodes utilizing lithium with less expensive mixed metal oxides and containing at least some manganese provide acceptable performance and cost a little over half as much as the current commercial lithium cobalt cathode systems. Materials utilized as cathodes in lithium-ion rechargeable cells according to the present invention exhibit increased discharge capacity, stable cycling performance, improved safety, and lowered impedance. The present invention could be incorporated into rechargeable lithium-ion batteries for electric vehicles, hybrid electric vehicles and portable electronic devices. The positive electrode active material for lithium-ion rechargeable battery of the present invention is represented by the general formula $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{A}_{\gamma}\text{O}_{2}$ wherein A is Mg, Zn, Al, Co, Ga, B, Zr, or Ti and $0 < x \le 0.2$, $0.1 \le \alpha \le 0.5$, $0.4 \le \beta \le 0.6$, $0 < \gamma \le 0.1$ having a layered structure.

[0010] It is therefore an object of the present invention to provide an improved positive active material with very high capacity for lithium-ion rechargeable battery which is stable at high voltage.

[0011] It is another object of the present invention to provide an improved method of manufacturing the positive active material having the above characteristics.

[0012] These and other objects, advantages and features of the invention together with the organization and manner of operation thereof will become apparent from the following detailed description when taken into conjunction with the accompanying drawings.

>

(ANL-IN-02-081)

DETAILED DESCRIPTION OF THE DRAWINGS

[0013] FIGURE 1 shows X-ray diffraction patterns (a) of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ calcined at 900°C and quenched into liquid nitrogen; (b) after then being cooled at 2°C/min; and (c) after then being cooled at 0.5°C/min (c);

[0014] FIGURE 2(a) shows charge/discharge curves of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ quenched into liquid nitrogen; FIGURE 2(b) shows charge/discharge curves of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ cooled at 2°C/min; and FIGURE 2(c) shows charge/discharge curves of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ and cooled at 0.5°C/min (c); and

[0015] FIGURE 3 shows the variation of discharge capacity with cycling number (a) of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ calcined at 900°C and then quenched into liquid nitrogen; and (b) after then being cooled at 2°C/min; and (c) after then being cooled at 0.5°C/min.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENTS

[0016] The present invention provides an improved positive active material with a very high capacity for use in an electrode, such as but not limited to a cathode in a lithium-ion rechargeable battery which is stable at high voltage. The present invention also relates to improved method of manufacturing the positive active material having the above characteristics. In one embodiment, the present invention relates to a material and method for making the same, wherein the material has the general formula $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{A}_{\gamma}\text{O}_{2}$ and further wherein A is Mg, Zn, Al, Co, Ga, B, Zr, or Ti and $0 < x \le 0.2, 0.1 \le \alpha \le 0.5, 0.4 \le \beta \le 0.6, 0 < \gamma \le 0.1$.

[0017] In one embodiment, a solid-state reaction method is employed to create the positive active material. In this embodiment, amounts of lithium hydroxide (or lithium carbonate), (Ni,Mn)-hydroxide, and A-hydroxide (or A oxide; A=Mg, Zn, Al, Co, Ga, B, Zr, Ti) are mixed in acetone using zirconia balls for about 12-24 hours to form a mixed powder. Next the positive electrode active material is manufactured by firing the mixture of lithium carbonate (or lithium hydroxide), (Ni,Mn)-hydroxide, and A-hydroxide (or A-

(ANL-IN-02-081)

oxide), preferably calcining the mixed powder at about 450~550°C for about 12-30 hours in air and then calcining the mixed powder at about 900-1100°C for about 10-24 hours either in an air or an nitrogen-oxygen atmosphere. This is followed by a rapid quenching into liquid nitrogen.

[0018] In another embodiment, an aqueous solution method is employed to create the positive active material. Lithium hydroxide, nickel hydroxide, manganese acetate, and A-hydroxide (or A-nitrate) are used as starting materials for the synthesis. Appropriate amounts of lithium hydroxide, nickel hydroxide, and A-hydroxide (or A-nitrate; A = Mg, Zn, Al, Co, Ga) are dissolved in distilled water whose pH is adjusted with nitric acid. An aqueous solution of manganese acetate is added to the solution to form a mixed solution. The the mixed solution is refluxed in a round bottom flask attached with a condenser at about 80°C for about 12-24 hours. Next, the mixed solution is evaporated in a rotary vacuum evaporator. The organic contents in the mixed solution are eliminated by heating at about 400°C for about 2 hours. The calcination is carried out in a temperature range of about 900-1100°C for about 10-24 hours in either an air or an nitrogen-oxygen atmosphere. This step is followed by a rapid quenching into liquid nitrogen. [0019] In yet another embodiment, a sol-gel method is employed to create the positive active material. Acetates of lithium, nickel, manganese, and A are employed as starting materials; and glycolic acid/tartaric acid solution is used as a chelating agent. Appropriate amounts of lithium acetate, nickel acetate, manganese acetate, A-acetate (or A-nitrate; A = Mg, Zn, Al, Co, Ga) are dissolved in distilled water. A glycolic/tartaric acid solution is adding for use as a chelating agent. The pH of the resulting solution is adjusted to about 7~8, preferably using ammonium hydroxide. The solution is continuously stirred and heated, preferably on a hot plate to form a gel precursor. The gel precursor is decomposed at about 450°C about for about 5hours in air to form a decomposed powder. The calcinations are preferably carried out in a temperature range from about 900°C to 1100°C for about 10-24 hours in air or nitrogen/oxygen mixture atmospheres. This is followed by a rapid quenching into liquid nitrogen.

(ANL-IN-02-081)

[0020] In one embodiment, the material in accordance with the principles of the present invention has a composition of $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{O}_2$, wherein x is between about 0 and 0.2, α is between about 0.2 and 0.5, and β is between about 0.5 and 0.6.

[0021] The material in accordance with the principles of the present invention, in another embodiment has a composition of $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{Mg}_{\gamma}\text{O}_2$, wherein x is between about 0 and 0.2, α is between about 0.1 and 0.5, β between about 0.4 and 0.6, and γ is between about 0.01 and about 0.1.

[0022] In yet another embodiment, present invention comprises a material composed of $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{Al}_{\gamma}\text{O}_2$, wherein x is between about 0 and 0.2, α is between about 0.15 and 0.5, β is between about 0.45 and 0.6, and γ is between about 0.01 and about 0.1.

[0023] In yet another embodiment, the material in accordance with the principles of the present invention has a composition of $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{Co}_{\gamma}\text{O}_{2}$, wherein x is between about 0 and 0.2, α is between about 0.15 and 0.5, β is between about 0.45 and 0.6, and γ is between about 0.01 and about 0.1.

[0024] The material in accordance with the principles of the present invention, in another embodiment, has a composition of $\text{Li}_{1+x}\text{Ni}_{\alpha}\text{Mn}_{\beta}\text{Ti}_{\gamma}\text{O}_2$, wherein x is between about 0 and 0.2, α is between about 0.2 and 0.5, the β is between about 0.4 and 0.6, and γ is between about 0.01 and about 0.1.

[0025] In one embodiment, the active material in accordance with the principles of the present invention may be used as a cathode. In accordance with the principles of the present invention, the active material may be used in a rechargeable lithium-ion battery.

[0026] In one example, an undoped system Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ was selected as an example of positive electrode active material of a high-energy lithium-ion rechargeable battery and the sol-gel method was selected as the manufacturing method. The sol-gel preparation of the lithium nickel manganese oxide was investigated by using lithium acetate, nickel acetate, and manganese acetate. Each starting material was dissolved in distilled water and added to glycolic/tartaric acid solution that was used as a chelating agent. The solution pH was adjusted to around 7 using ammonium hydroxide. The entire

(ANL-IN-02-081)

process was conducted under continuous stirring and heating on a hot plate. The resulting gel precursor was decomposed at 450°C for 5 hours in air. The decomposed powders were then fired at 900°C for 24 hours in air atmosphere. Finally, the calcined powders were quenched rapidly into liquid nitrogen. To demonstrate the effectiveness of the rapid quenching, materials which were slowly cooled (for example, at a rate of about 2°C/min and 0.5°C/min) were also manufactured.

[0027] X-ray diffraction analysis of the obtained Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂, as shown in FIGS. 1(a)-1(c) show an ordered rock-salt structure (crystallographic structure of $R\bar{3}m$) with extra peaks at 20~25° (marked with arrows) (shown at FIG. 1(a)). These are generally attributed to cation ordering (Li/Mn and/or Li/Ni) in the transition metal layers. The materials cooled at slower rates, shown at FIG. 1(b) and FIG. 1(c), show many extra peaks (marked with circles) over the whole 20 range, which is a typical character of monoclinic symmetry.

[0028] In order to study the electrochemical properties of Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂ powder obtained as described above, coin type cells were made and galvanostatically cycled. The cathode mixture consisted of 80 wt.% active material, 10 wt.% carbon, and 10 wt.% Polyvinylidene fluoride (PVDF) as a binder. The negative electrode was metallic lithium and the electrolyte was 1 M LiPF₆ in a 1:1 mixture of ethylene carbonate (EC)/diethyl carbonate (DEC). The galvanostatic cycling was performed in the voltage range of 2.0 — 4.6V at a current density of 0.1 mA/cm².

[0029] FIGS. 2(a)-2(c) and FIG. 3 show the charge/discharge characteristics and cycling performance of the obtained Li(Li_{0.17}Ni_{0.25}Mn_{0.58})O₂. The obtained materials exhibited very high discharge capacity of 210 mAh/g with excellent cycling performance, whereas the samples cooled at slow rates delivered discharge capacities only of 130 mAh/g (2°C/min cooling) and 85 mAh/g (0.5°C/min).

[0030] While several embodiments have been shown and described in this application, it is understood that changes and modifications can be made to the invention without departing from the invention's broader aspects.